A Facile One-Step Synthesis of TiO₂/Graphene Composites for Photodegradation of Methyl Orange

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ABSTRACT

 TiO_2 /graphene composite photocatalysts have been prepared by a simple liquid phase deposition method using titanium tetrafluoride and electron beam (EB) irradiation-pretreated graphene as the raw materials. The products were characterized by X-ray diffraction, transmission electron microscopy, X-ray photoelectron spectroscopy, and thermogravimetric analysis. The effects of varying the synthesis parameters such as graphene content, concentration of titanium tetrafluoride solution and irradiation dose were investigated. It was found that the preparation conditions had a significant effect on the structure and properties of the final products. The irradiated graphene was covered with petal-like anatase TiO_2 nanoparticles, which were more uniform and smaller in size than those in products synthesized without EB irradiation-pretreated graphene. The photocatalytic activities of the products were evaluated using the photocatalytic degradation of methyl orange as a probe reaction. The results showed that the products synthesized using EB irradiation-pretreated graphene exhibited higher photocatalytic activities than those using graphene without EB irradiation pretreated.

KEYWORDS

TiO₂/graphene composites, electron beam (EB) irradiation, photocatalytic degradation, methyl orange

1. Introduction

Titanium dioxide (TiO₂) is an important semiconductor, which is used in environmentally benign processes because of its advantageous properties such as nontoxicity, photo- and chemical stability, water insolubility under most conditions and low production cost [1, 2]. However, it is well known that the morphology and size of TiO₂ particles play very crucial roles in its properties and applications [1, 3]. To date, many synthesis methods have been developed in an attempt to improve the photocatalytic activity of powdered TiO₂ catalysts, including noble metal deposition, ion doping [4–7], and adding a co-adsorbent to TiO₂ [8–12]. Among these materials, multi-walled carbon nanotube (MWCNT)–TiO₂ composites have been studied extensively, and have proved to be very effective in improving the photocatalytic activity of TiO₂ [13–15].

Graphene, with its unique structure of one-atomthick planar sheets of sp²-bonded carbon atoms closely packed in a honeycomb crystal lattice, has attracted considerable attention due to its extraordinary mechanical and electronic properties [16–20]. Graphene

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can be viewed as split carbon nanotubes (CNTs), suggesting it has great potential to enhance the photocatalytic activity of catalysts like TiO₂ [13, 21]. Compared with CNTs, graphene has many advantages including high surface area (a theoretical value of 2630 m²/g) [22] and good interfacial contact with adsorbates. Therefore, it is desirable to explore simple and effective approaches for synthesizing graphenebased composites and expand their applications. Very recently, there have been a few reports of the synthesis of TiO₂/graphene composites. For instance, TiO₂ particles and graphene oxide (GO) colloids have been mixed ultrasonically, followed by ultraviolet (UV)-assisted photocatalytic reduction of GO to yield TiO₂/graphene composites [23]. In another example, TiO₂/graphene composite materials were prepared by self-assembly of nanocrystalline TiO₂ grown in situ on graphene by a one-step approach with the assistance of an anionic surfactant [24]. There was another report on the use of titania P25-graphene composites as photocatalysts [25], in which the morphology of the composites remains to be optimized in order to achieve higher photocatalytic activity. Hence, it is still a great challenge to prepare TiO₂/graphene composites with the appropriate crystal structure to achieve high photocatalytic activity.

In this work, we describe a liquid phase deposition method for the preparation of TiO₂/graphene composites. The effect of electron beam (EB) irradiation of the graphene on the structure and properties of the materials was studied. As illustrated in Fig. 1, EB irradiation can arouse oxygenated groups on graphene



Figure 1 Diagram of the chemical route to the TiO_2 /graphene composites

surface. The carbon oxygen bands excitated by EB irradiation on graphene surface can be considered as defect sites, which could increase the affinity of graphene to Ti⁴⁺ through electrostatic force, thus providing more growing spots for TiO₂ crystal. At the same time, the excited oxygenated groups were not very stable and could be partially reduced during the growth of TiO₂. The optimal experimental conditions were established by varying the reaction conditions. The photochemical degradation of methyl orange was employed as a probe reaction to determine the photocatalytic activity of the composites and their activities were compared with that of a mixture of P25 and graphene.

2. Experimental

2.1 Synthesis of TiO₂/graphene composites

Prior to the synthesis of TiO_2 /graphene composites, GO was synthesized from natural graphite powder by a modified Hummers' method [26]. Then, assynthesized GO was calcined at 500 °C for 2 h in a nitrogen atmosphere and the resulting powder was collected for use in the next step.

Typically, 20 mg of graphene prepared in the previous steps was added to 40 mL of 0.04 mol/L TiF₄ (from Acros Organics, Belgium) solution in a 200 mL glass beaker. The glass beaker was sealed with Parafilm and placed in an ultrasonic water bath for 50 min to ensure good dispersion of the graphene. The resulting mixture was then heated in an electric oven at 60 °C for 20 h. After reaction, the parafilm was found broken, due to the build-up pressure upon the reaction. The mixture was washed several times with deionized water and ethanol solution, and dried in an oven at 60 °C to obtain the final products.

Following the same procedure, a series of composite catalysts were prepared by varying the content of graphene and concentration of the TiF_4 solution.

In an attempt to improve the photocatalytic activity, EB irradiation was employed to irradiate the graphene before reaction. In a typical procedure, graphene was dispersed in aqueous solution, and then the mixture was placed in a polythene bag and irradiated under the electron beam of an accelerator with a dose of 2.1×10^5 Gy and a current of 7.0 mA. With all other experimental conditions being kept the same, experiments were also carried out by varying the dose of EB irradiation on graphene.

2.2 Catalyst characterization

Samples were characterized by X-ray diffraction (XRD, Rigaku D/max-2550, Cu Kα radiation), ultraviolet and visible spectroscopy (UV-vis, U-3010), transmission electron microscopy (TEM, JSM-2010F) and highresolution transmission electron microscopy (HRTEM, JSM-2010F). Raman spectra of samples were recorded using a Renishaw inVia system with an excitation wavelength of 785 nm and a charge-coupled device detector. Thermogravimetric (TG) and differential thermogravimetric (DTG) analysis was carried out with a STA 409 PC/4/H Luxx system using a heating rate of 10 °C/min in a flow of air. N2 adsorption-desorption isotherms were measured on a QUADRASORB SI Surface Area and Pore Size Analyzer at 77 K using the volumetric method, after the as-prepared samples were annealed at 200 °C. The Brunauer-Emmett-Teller (BET) specific surface area was calculated by using adsorption data. Atomic force microscopy (AFM) images were obtained by using a SPM-9600 instrument (Shimadzu, Japan). The substrates used for AFM imaging were freshly cleaved mica substrates. X-ray photoelectron spectroscopy (XPS) measurements were performed on an AXIS Ultra DLD (Kratos, USA) using monochromatic Al K α X-ray source (anode HT = 15 kV) operating at a vacuum better than 10-7 Pa. Kratos Vision v. 2.2 software was utilized to analyze and deconvolute the XPS peaks.

2.3 Photodegradation reactions

The photocatalytic activity of TiO₂/graphene composites was evaluated using methyl orange degradation in aqueous solution under UV light irradiation. Experiments were carried out in an SGY-IB photochemical reactor. Typically, 20 mg of as-prepared TiO₂/graphene composite catalyst was added to 50 mL of 0.025 g/L methyl orange solution in a 100 mL beaker. The mixture was then placed in an ultrasonic water bath for 30 min to ensure good dispersion of the catalyst, followed by stirring in the dark at ambient

temperature for 1 h to achieve adsorption-desorption equilibrium for methyl orange. After reaching equilibrium, 3-5 mL of suspension was extracted in order to determine the initial concentration of methyl orange solution, which was recorded as the base concentration C_0 . The remaining mixture was transferred to a 50 mL quartz tube and illuminated with a 300 W Hg lamp. In the following 60 min, 3–5 mL aliquots of the suspension were removed every 10 min. The suspension was centrifuged immediately to separate any suspended solid. The UV-vis spectra of the supernatant were recorded using a Hitachi U-3010 UV-vis spectrophotometer to determine the concentration of methyl orange at each time, which is denoted as C_{t} . For comparison, a mixture of P25 and irradiated/unirradiated graphene (the amount of TiO₂ was kept the same as in the composites) and bare graphene (20 mg) were employed under the same conditions.

3. Results and discussion

3.1 Characterization of GO and graphene sheets

As shown in Fig. S-1 in the ESM, the thickness of the GO layer was 1.32 nm, larger than the typical thickness of single-layer GO (~0.8 nm), but still smaller than the typical thickness of a bilayer graphene, indicating that single-layer graphene accounted for the major part of the prepared materials [27]. In addition, as calcination at 500 °C in an N₂ atmosphere does not result in restacking of GO sheets, it can be deduced that the graphene used in this study is mostly single-layer graphene.

XPS was employed to study the chemical states of the prepared samples. The peak deconvolution of the C (1s) XPS core level of GO sheets and graphene sheets reduced by annealing at 500 °C are shown in Fig. S-2 in the ESM. Peaks with binding energies of 284.6 eV and 285.4 eV can be attributed to the C–C and C (CH₂) bonds, respectively. The deconvoluted peaks centered at binding energies of 286.2 and 288.1 eV can be assigned to C–OH and C=O functional groups, respectively [27–30]. Before the annealing treatment, considerable contributions from the oxygen-containing functional groups were observed. After annealing



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treatment, however, the areas of these peaks decreased considerably due to the reduction process. The peak area ratios of the C–OH and C=O bonds to the C–C bonds are summarized in Table S-1 in the ESM. The relative concentrations of the C–OH and C=O bonds showed decreases of 73% and 78% from the corresponding concentrations for the GO sheets. This indicates that an effective, but not complete, chemical reduction of the GO sheets to graphene has occurred. The residual small amount of oxygen-containing bonds should not significantly affect the properties of the graphene, however.

3.2 Effect of the graphene content

The effect of graphene on the catalytic activity of the prepared composite catalyst was studied by varying the content of graphene while keeping the other conditions constant. Figure 2 presents the XRD patterns of the catalysts when the graphene dose was 10 (a), 20 (b), 30 (c), and 40 mg (d). It can be observed that a pure anatase TiO₂ phase (JCPDS No. 21-1272) was obtained for a graphene content of 20 mg. For other graphene contents, additional peaks from impurities such as TiOF and TiOF₂ appeared (marked with symbols "*" and "#", respectively). As illustrated in the experimental section, a parafilm was used to seal the beaker during the reaction. And the as-prepared samples under our experimental condition have more uniform morphology compared with that prepared under completely closed or open condition (data not shown), corresponding to the present literature report [15]. The TiO₂/graphene composites were obtained from the hydrolysis of TiF₄ in the presence of aqueous dispersions of graphene, as shown in eq 1.

$$TiF_4$$
 + graphene + $2H_2O \cdot TiO_2$ /graphene(s) + $4HF$ (1)

It has been reported that the hydrolysis reaction to form TiO_2 rely on the content of the graphene [31]. When 10 mg graphene was suspended in the system, the reaction rate was slow, with the reaction progressing, a part of water evaporated and large amount of residual TiF_4 would form high concentration TiF_4 solution, leading to formation of TiOF and $TiOF_2$ impurities. While at high graphene content (30 mg, 40 mg), the initial reaction rate was so fast that the parafilm was



Figure 2 XRD patterns of TiO_2 /graphene composites synthesized with different graphene contents (a) 10 mg, (b) 20 mg, (c) 30 mg, and (d) 40 mg

soon broken by generated HF gas, then the water would evaporated fast and high concentration TiF_4 solution would form, also leading to TiOF and TiOF₂ impurities. In addition, it is worth mentioning that the TiOF and TiOF₂ impurities were transformed to anatase TiO₂ after calcination at 300 °C for 2 h. However, since calcination would increase the particle size of TiO₂ and hence decrease its photocatalytic activity, all the samples used in our work were obtained without calcination.

Figure 3 shows the Raman spectra of $TiO_2/graphene$ composites prepared with 20 mg of graphene and 0.04 mol/L TiF₄ solution. A well resolved TiO_2 Raman



Figure 3 Raman spectra of the TiO_2 /graphene composite prepared with 20 mg of graphene and 0.04 mol/L TiF_4 solution. The inset shows Raman spectra (800 to 2000 cm⁻¹) of (a) the TiO_2 /graphene composite and (b) pure graphene

peak is clearly seen at about 153 cm⁻¹, which is attributed to the main E_g anatase vibration mode. Furthermore, vibration peaks at 400 cm⁻¹ (B_{1g}), 517 cm⁻¹ (A_{1g}) and 640 cm⁻¹ (E_g) are also characteristic of anatase TiO₂ [32].

The Raman spectra are therefore consistent with the XRD results shown in Fig. 2(b). Additionally, two peaks at about 1320 cm⁻¹ (D band) and 1594 cm⁻¹ (G band) are observed in the spectra (see the inset in Fig. 3), which can be attributed to the graphene substrate. The intensity ratios (I_D/I_G) for bare graphene and the TiO₂/graphene composite were 2.31 and 2.02, respectively. The decrease in the D/G intensity ratio for the TiO₂/graphene composite suggests that a decrease in the average size of the sp² domains of C atoms took place during the growth of anatase crystallites [33].

Figure 4 displays the TG and DTG curves of asprepared graphene and the $TiO_2/graphene$ composite prepared with 20 mg of graphene and 0.04 mol/L TiF_4 solution. As shown in Fig. 4(a), the pure graphene

100 0.190 0.0 Weight percentage (%) 80 TG -0.1 70 -0.2 60 G (%/min) DTG 50 0.3 40 -0.4 30 DI -0.5 20 10 -0.6 0 -0.7 100 200 300 400 500 600 700 800 900 Temperature (°C) (a) 100 0.1 90 0.0 80 Weight percentage (%) DTG -0.1 70 -0.2(%/min 60. TG 50 -0.3 40 DTG -0.4 30 -0.5 20. -0.6 10 0 -0.7100 200 300 400 500 600 700 800 900 Temperature (°C) (b)

Figure 4 TG and DTG curves of (a) pure graphene and (b) the TiO_2 /graphene composite prepared with 20 mg of graphene and 0.04 mol/L TiF_4 solution

decomposed completely on heatingin air below 900 °C. The DTG curve exhibits strong bands centered at about 687 °C, which correspond to the gasification of graphene. In Fig. 4(b), the TG curve of the TiO₂/graphene composite shows a small mass loss of 12% and a larger mass loss of 25%, which can be attributed to water removal and structural water evolution, and graphene gasification, respectively. In the DTG curve, the temperature at which the maximum rate of graphene gasification occurs is about 685 °C, which is consistent with that of pure graphene.

In Fig. 5(a), C (1s) for TiO_2 /graphene composites prepared with 20 mg of graphene and 0.04 mol/L TiF_4



Figure 5 Peak deconvolution of C (1s) and Ti (2p) XPS core levels of (a) a TiO₂/graphene composite prepared with 20 mg of graphene and 0.04 mol/L TiF₄ solution and (b) the TiO₂/graphene composite synthesized with 2.1×10^5 Gy irradiation



solution showed similar peak deconvolution results with that of annealing graphene. But as displayed in Table 1, a slightly increase in the peak area (*A*) ratios of the oxygen-containing bonds to the CC bonds was observed compared with that of bare graphene sheets. The reason may be that upon growth of TiO₂ on graphene surface, the concentration of the CC bonds may decrease slightly. The XPS spectrum at the Ti (2p) binding energy region was also present in Fig. 5 (b). The two bands located at binding energies of 465.2 and 459.5 eV were assigned to the Ti (2p_{1/2}) and Ti (2p_{3/2}) spin-orbital splitting photoelectrons in the Ti⁴⁺ chemical state, respectively. A 5.7 eV peak-to-peak separation indicated the formation of TiO₂ on graphene surface [34, 35].

3.3 Effect of the TiF₄ concentration

The concentration of TiF_4 was varied while the content of graphene was kept constant at 20 mg. Figure 6 shows XRD patterns of the samples when the TiF_4 concentration was 0.02 (a), 0.04 (b), 0.06 (c), and 0.08 mol/L (d). For a concentration of TiF_4 of 0.08 mol/L, the sample

Table 1 The peak area (A) ratios of the oxygen-containing bonds to the C–C bonds, for composites with and without irradiation

Irradiation dose	0 Gy	$2.1 \times 10^5 \text{ Gy}$
$A_{\rm COH}/A_{\rm CC}$	0.19	0.21
$A_{\rm CO}/A_{\rm CC}$	0.04	0.05



Figure 6 XRD patterns of TiO_2 /graphene composites synthesized with different TiF_4 concentrations of (a) 0.02 mol/L, (b) 0.04 mol/L, (c) 0.06 mol/L, and (d) 0.08 mol/L

appeared to contain TiOF and TiOF₂ impurities. The characteristic peaks of TiOF and TiOF₂ are marked with the symbols "*" and "#", respectively. As shown in the TEM images (Fig. 7), the growth of petal-like TiO₂ on graphene increases gradually with the increasing TiF₄ concentration.

Figure 8 presents the results of photodegradation reactions of methyl orange solution. As the pure TiO_2 nanoparticles prepared using the same procedure in



Figure 7 TEM images of TiO₂/graphene composites synthesized with different TiF₄ concentrations of (a) 0.02 mol/L, (b) 0.04 mol/L, (c) 0.06 mol/L, and (d) 0.08 mol/L



Figure 8 Conversion of methyl orange in photodegradation over TiO_2 /graphene composites synthesized with different TiF_4 concentration, together with the results for a mixture of P25 and graphene, and bare unirradiated graphene for comparison

the absence of graphene tended to agglomerate into bulk and exhibited very low photocatalytic efficiency, P25 (a commercial photocatalyst) and unirradiated graphene were used as comparison materials to evaluate photocatalytic activity of TiO₂/graphene composites. As shown in Fig. 8, bare unirradiated graphene did not demonstrate any photocatalytic activity. The TiO₂/graphene composite catalyst with the best performance appeared to be the one synthesized with 0.04 mol/L TiF₄ solution. The conversion of methyl orange was about 85% after UV light irradiation for 60 min, which is much higher than that of a mixture of P25 and graphene.

3.4 Effect of the irradiation dose

EB irradiation was used for pretreatment of graphene in our work. Various dosages of EB irradiation were employed whilst keeping the other conditions unchanged (including a content of graphene of 20 mg and a TiF₄ concentration of 0.04 mol/L). Figure 9 shows the XRD patterns of as-prepared TiO₂/graphene composites when the irradiation dose was 7×10^4 (a), 1.4×10^5 (b), 2.1×10^5 (c), and 2.8×10^5 Gy (d). These XRD patterns appear identical to those in Fig. 6, indicating that EB irradiation did not alter the TiO₂ anatase structure. TEM images of the corresponding samples are shown in Fig. 10. The TiO₂ cluster size



Figure 9 XRD patterns of TiO₂/graphene composites synthesized with different irradiation doses of (a) 7×10^4 Gy, (b) 1.4×10^5 Gy, (c) 2.1×10^5 Gy, and (d) 2.8×10^5 Gy



Figure 10 TEM images of TiO₂/graphene composites synthesized with different irradiation doses of (a) 7×10^4 Gy, (b) 1.4×10^5 Gy, (c) 2.1×10^5 Gy, and (d) 2.8×10^5 Gy

decreased when the EB irradiation dose was increased from 7×10^4 to 2.1×10^5 Gy. However, when the EB irradiation dose was increased to 2.8×10^5 Gy, the TiO₂ crystal cluster size increased compared to that of 2.1×10^5 Gy. To further explore the effect of EB irradiation, HRTEM was employed for sample characterization. As shown in Fig. 11, in both cases with and without EB irradiation, the petal-like structures of TiO₂ grown on graphene are composed of numerous tiny TiO₂ crystallites. However, the crystal cluster size of the samples prepared with EB irradiation pretreatment (Figs. 11(c) and 11(d)) appear to be smaller and more uniformly distributed than those in samples prepared without EB irradiation (Figs. 11(a) and 11(b)). In addition, N₂ adsorption-desorption isotherms were also employed to determine the specific surface area of prepared samples. The calculated BET surface areas of materials produced with and without EB irradiation pretreatment were 175.7 and 51.7 m^2/g_r respectively. The corresponding total pore volumes were 0.28 and 0.11 cm3/g, respectively. These results show that 2.1×10^5 Gy of irradiation can reduce crystallite cluster size and increase BET surface area of the TiO₂/graphene composites, both of which changes which should enhance the photocatalytic activity of the materials in the degradation of methyl orange solution.





Figure 11 HRTEM images of TiO₂/graphene composites synthesized (a, b) without EB irradiation pretreatment, and (c, d) with EB irradiation pretreatment of 2.1×10^5 Gy

To explain the effect of EB irradiation, XPS spectra and peak area (A) ratios of the oxygen-containing bonds to the CC bonds for graphene irradiated with different irradiation dose were present in Fig. S-3 and Table S-2 in the ESM, respectively. The results showed that when the EB irradiation dose increased from 7×10^4 to 2.1×10^5 Gy, the peak area ratios of the C–OH and C=O bonds to the C-C bonds increased gradually. While when the dose was increased to 2.8×10^5 Gy, ACOH, AOH as well as Acc decreased greatly compared with that of unirradiated graphene, indicating the damage in lattice structure of graphene. The results are consistent with present reports [36, 37]. The carbon oxygen bands excitated by EB irradiation on graphene surface can be considered as defect sites, which could increase the affinity of graphene to Ti⁴⁺ through electrostatic force, thus providing more growing spots for TiO₂ crystal. With the growing of TiO₂ crystal, the steric hindrance could affect the morphology of TiO₂ growing on the surface of graphene. And more defect sites lead to smaller TiO₂ cluster size. Herein, the TiO₂ cluster size decreased when the EB irradiation dose was increased from 7×10^4 to 2.1×10^5 Gy.

The deconvoluted C (1s) and Ti (2p) XPS spectra of the TiO₂/graphene composite synthesized with 2.1 × 10^5 Gy irradiation are shown in Figs. 5(a) and 5(b). They are consistent with that of composites synthesized without irradiation. Seen from Table 1, only a slightly increase in ratios of the C–OH and C=O bonds was observed, which indicated that the excited oxygenated groups by EB irradiation were not very stable and could be partially reduced during the growth of TiO₂. Herein, though the EB irradiation would increase the oxygenated groups on graphene, the properties of graphene in the prepared TiO₂/graphene composites would not be greatly affected.

Figure 12 displays the catalytic performance in photodegradation reactions of the catalysts prepared with different doses of EB irradiation. Bare 2.1×10^5 Gy EB irradiated graphene did not exhibit any photocatalytic activity, and the mixture of irradiated graphene and P25 demonstrated no obvious increase in photocatalytic efficiency in comparison with that of the mixture without irradiation. The TiO₂/graphene composite prepared with 2.1×10^5 Gy of EB irradiation showed the best catalytic performance. This indicates that EB irradiation pretreatment can lead to a beneficial morphological change in the catalyst resulting in an enhancement of photocatalytic activity. In particular,



Figure 12 Conversion of methyl orange in photodegradation using TiO₂/graphene composite catalysts synthesized with different irradiation doses, with the results for a mixture of P25 and 2.1 × 10^5 Gy irradiated graphene, and bare irradiated graphene for comparison

it appeared that the photocatalytic activity of samples increased with increasing EB irradiation dosage in the order $2.1 \times 10^5 > 1.4 \times 10^5 > 2.8 \times 10^5 > 7 \times 10^4$ Gy.

Based on the comparison of TEM images, it can be concluded that the photocatalytic activity of the composites increased with decreasing TiO_2 crystal cluster size. The reason may be that decreasing the TiO_2 crystal cluster size results in an increase in contact area between the catalyst and organic molecules, which favors the degradation of methyl orange.

4. Conclusions

In summary, the TiO₂/graphene composite catalysts were prepared using a facial one-stepn easy and environmentally benign metal. Key synthesis parameters such as the graphene content, concentration of TiF₄ solution and EB irradiation dose were optimized in our experiments. The TEM results showed that EB irradiation pretreatment of graphene is an effective way to produce uniform and small size TiO₂ nanoparticles on graphene. More importantly, the photodegradation results confirmed that EB irradiation pretreatment of graphene could significantly enhance the photocatalytic activity of TiO₂ in the degradation of methyl orange. Certainly, however, further improvements in photocatalytic performance of TiO₂/graphene composites are needed in order to develop practical applications in the environmental field.

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Electronic Supplementary Material: Supplementary material (AFM image of GO, XPS spectra of GO, unirradiated graphene and irradiated grapheme, Table of peak area (A) ratios of the oxygen-containing bonds

to the CC bonds for GO and EB irradiated graphene) is available in the online version of this article at http://dx.doi.org/10.1007/s12274-010-0079-4.

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Table of contents



TiO₂/graphene composite photocatalysts have been prepared by a simple liquid phase deposition method using titanium tetrafluoride and electron beam (EB) irradiation-pretreated graphene as the raw materials, and their photocatalytic properties studied.



Electronic Supplementary Material

A Facile One-Step Synthesis of TiO₂/Graphene Composites for Photodegradation of Methyl Orange

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Figure S-1 AFM image of as-prepared GO sheets







Figure S-2 Peak deconvolution of the C (1s) XPS core level of (a) GO sheets and (b) graphene sheets reduced by annealing at 500 °C



Figure S-3 Peak deconvolution of the C (1s) XPS core level of graphene irradiated with different irradiation doses of (a) 7×10^4 Gy, (b) 1.4×10^5 Gy, (c) 2.1×10^5 Gy, and (d) 2.8×10^5 Gy

Table S-1 The peak area (A) ratios of the oxygen-containing bonds to the C–C bonds, for graphene sheets before and after annealing

Sample	Graphene oxide sheets	After annealing	
$A_{\rm COH}/A_{\rm CC}$	0.67	0.18	
$A_{\rm CO}/A_{\rm CC}$	0.09	0.02	

Table S-2 The peak area (A) ratios of the oxygen-containing bonds to the C–C bonds, for graphene irradiated with different irradiation doses

Irradiation dose	$7 imes 10^4 \ Gy$	$1.4 \times 10^5 \text{ Gy}$	$2.1 \times 10^5 \text{ Gy}$	$2.8\times 10^5~Gy$
$A_{\rm COH}/A_{\rm CC}$	0.19	0.20	0.24	0.19
$A_{\rm CO}/A_{\rm CC}$	0.02	0.03	0.06	0.03

